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# Study of sequential abiotic and biotic degradation of styrene butadiene rubber

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## Keywords

Photo-ageing; Thermo-ageing; Biodegradation, *Rhodococcus ruber* ATCC29672; *Gordonia  
polyisoprenivorans* VH2

## Abstract

Styrene butadiene rubber is one of the main constituents of tire tread. During tire life, the tread material undergoes different stresses that impact its structure and chemical composition. Wear particles are then released into the environment as weathered material. To understand their fate, it is important to start with a better characterization of abiotic and biotic degradation of the elastomer material. A multi-disciplinary approach was implemented to study the photo- and thermo- degradation of non-vulcanized SBR films containing 15 w% styrene as well as their potential biodegradation by *Rhodococcus ruber* and *Gordonia polyisoprenivorans* bacterial strains. Each ageing process leads to crosslinking reactions, much surface oxidation of the films and the production of hundreds of short

25 chain compounds. These degradation products present a high level of unsaturation and oxidation and  
26 can be released into water to become potential substrates for microorganisms. Both strains were able  
27 to degrade from 0.2 to 1.2 % (% ThOD) of the aged SBR film after 30-day incubation while no  
28 biodegradation was observed on the pristine material. A 25-75 % decrease in the signal intensity of  
29 water extractable compounds was observed, suggesting that biomass production was linked to the  
30 consumption of low-molecular-weight degradation products. These results evidence the positive  
31 impact of abiotic degradation on the biodegradation process of styrene butadiene rubber.

32

### 33 **1. Introduction**

34 Styrene butadiene rubber (SBR) is the main elastomer in the formulation of car tire treads and one of  
35 their main components (30-70 wt%) (Babbit, 2010; Baumann and Ismeier, 1998). During the use  
36 phase on the road, the tire undergoes various physico-chemical stresses such as mechanical shear  
37 stress that leads to the degradation and release of SBR into the environment as tire and road wear  
38 particles (TRWP). Once generated, these particles are transported into various environmental  
39 compartments (air, soil, runoff, estuaries, sediments) (Baensch-Baltruschat et al., 2021; Goßmann et  
40 al., 2021; Klöckner et al., 2021, 2020; Leads and Weinstein, 2019; Panko et al., 2013; Parker-Jurd et  
41 al., 2021; Rauert et al., 2022; Unice et al., 2012), where further SBR degradation processes can occur,  
42 such as photo and thermooxidation and/or microbial degradation (Wagner et al., 2022). Large masses  
43 of TRWP released into the environment (0.2 to 5.5 kg/inhabitant/year) raise concerns about their  
44 potential impact on the ecosystems and questions about their persistence and fate (Baensch-  
45 Baltruschat et al., 2020). A thorough characterization of the complex chemical composition of TRWP  
46 is thus necessary to better understand the transformation of these particles. Such a task is challenging  
47 as the abiotic and biotic transformation of the elastomers and formulating agents in tire tread  
48 composition must be considered. While several studies recently identified the transformation products  
49 (*e.g.* 6PPD quinone, benzothiazole) of some tire curing and protective agents resulting from abiotic  
50 ageing (Calarnou et al., 2023; Fohet et al., 2023; Hu et al., 2022; Seiwert et al., 2022; Thomas et al.,  
51 2022; Unice et al., 2015), literature concerning the fate of elastomers is quite limited and often based

52 on the study of model materials. The effect of light on the ageing of vulcanized rubber was first shown  
53 in 1926 (Jecusco, 1926). Much research work subsequently studied the effects of UV, temperature or  
54 ozone on non-vulcanized 1,4-cis-isoprene rubber (IR), butadiene rubber (BR) and SBR (Adam et al.,  
55 1989a; Chandra, 1983; Lacoste et al., 1994) and vulcanized SBR and natural rubber (NR) (Osswald et  
56 al., 2019; Thomas et al., 2022) to predict their impacts on rubber/tire properties and performance. The  
57 mechanisms of photo and thermo-degradation of SBR were proposed for the first time in 1989 by  
58 Adam et al. (1989a). In the presence of oxygen, UV radiation and temperature induce significant  
59 modifications in the molecular network of SBR through cross-linking and cleavage reactions, resulting  
60 in the formation of oxidized degradation products with a reduced molecular weight (Chandra, 1983).  
61 The photo and thermooxidized SBR present various new functional groups including hydroperoxides,  
62  $\alpha,\beta$ -unsaturated ketones and saturated ketones, acids, alcohols and epoxides (Adam et al., 1989a). In  
63 ageing studies, the occurrence of such degradation products in polymers is conventionally measured  
64 by FTIR spectroscopy (Wilhelm and Gardette, 1994) and combined with physico-chemical  
65 characterization to evaluate the loss of material properties. Nevertheless, the formation rate of  
66 degradation products, their characterization and their leaching potential into the environment remain  
67 understudied. Recently, Thomas et al. (2022) proposed a list of possible transformation products of  
68 elastomers resulting from thermal or photo-ageing of tire cryogrinds. Oxidation products such as  
69 carboxylic acids, ketones and epoxides were detected in THF extracts by GC/MS. These components  
70 as well as the residual solid matrix of the elastomer (crosslinked and non-crosslinked) must be  
71 considered as potential carbon sources for microorganisms.

72 The current literature on synthetic elastomer biodegradation is mainly focused on biotechnological  
73 approaches for the purpose of rubber waste recycling. In that context, several studies used either actual  
74 tire tread materials (Tsuchii et al., 1996; Tsuchii and Tokiwa, 1999), ground tire rubber (Cui et al.,  
75 2016; Li et al., 2012, 2013) or model vulcanized rubber (Aboelkheir et al., 2019; Hu et al., 2016)  
76 where the elastomer and all the additives can potentially serve as substrates for microorganisms.  
77 Aboelkheir et al. (2019) showed surface modifications and suggested carbon and sulfur bond  
78 cleavages of vulcanized SBR in the presence of *Bacillus subtilis*, *Pseudomonas aeruginosa* and

79 *Streptomyces* sp. Other strains, including *Gordonia* spp, *Sphingomonas*, *Acidithiobacillus* and  
80 *Acinetobacter* spp., were also studied for their abilities to devulcanize SBR by the cleavage of the  
81 sulfur bonds (Cui et al., 2016; Hu et al., 2016; Li et al., 2012, 2013). Conversely, studies on pure SBR  
82 or BR are very scarce (Nielsen et al., 2023; Williams, 1982). In 1984, Tsuchii et al. isolated a  
83 *Moraxella* strain from soil capable of degrading more than 40 % of 1,4-BR with a molecular weight  
84 (Mn) under 2,500 within 5 days.

85 Besides SBR and BR, many studies focused instead on the biodegradation of natural or synthetic IR,  
86 another elastomer entering tire formulation and other manufactured products (*e.g.* gloves, hoses,  
87 gaskets, conveyor belts). IR biodegradation was demonstrated by several pure microbial strains  
88 (Andler et al., 2022; Berekaa, 2006; Berekaa et al., 2000; Borel et al., 1982; Cheng et al., 2023;  
89 Ibrahim and El-ameen, 2013; Linos et al., 2000; Roy et al., 2006; Tsuchii et al., 1979) or by a  
90 consortium (Nguyen et al., 2020). Its metabolic pathways were particularly investigated for the  
91 bacteria *Gordonia polyisoprenivorans* VH2 (Hiessl et al., 2012), *Streptomyces* sp. K30 (Rose et al.,  
92 2005) and *Xanthomonas* sp. 35Y (Birke et al., 2017; Sharma et al., 2018). It was shown that the oligo-  
93 isoprenoids formed, resulting from the cleavage of the double bonds of the isoprene units, can be bio-  
94 assimilated by microorganisms (Bode et al., 2000; Hiessl et al., 2012).

95 In the environment, abiotic and biotic degradation processes occur simultaneously or consecutively.  
96 The impact of abiotic transformations of rubber on biodegradation has never been considered to our  
97 knowledge. In 1980, Cadle and Williams studied the environmental degradation of tire wear particles  
98 (TWP) in roadside soil and suggested that the elastomer degradation was mainly due to atmospheric  
99 oxidation rather than microbial attack (Cadle and Williams, 1980). Biodegradation of SBR can occur  
100 in environmentally aged roadway particles (see our recent work Calarnou et al., 2023). Herein, *R.*  
101 *ruber* ATCC 29672 was found to degrade 12 % of the SBR extractable fraction from TRWP within 14  
102 days. Photo and thermooxidation of polymers (polyethylene, polypropylene) have already been  
103 reported to produce short oxidized chains that are subsequently used by microorganisms as carbon  
104 substrates (Albertsson and Karlsson, 1990; Bonhomme et al., 2003; Corti et al., 2010; Eyheraguibel et  
105 al., 2017; Fontanella et al., 2013; Koutny et al., 2006; Rodrigo et al., 2021). From an environmental  
106 standpoint, biotic and abiotic degradations control the fate of chemicals. Therefore, assessing

107 biodegradation of such polymers, generally considered difficult to (bio)degrade, should integrate their  
108 prior abiotic chemical transformations and the variety of degradation carbon sources they can  
109 generate.

110 The present study continues our work on the biodegradation of roadway particles containing TRWP  
111 (Calarnou et al., 2023) that characterized the various potential carbon sources present (elastomers,  
112 plasticizers, curing agents, antioxidants and their degradation products) describing their  
113 biodegradation. The complexity observed in TRWP hindered understanding (bio)degradation of each  
114 compound, especially of the elastomer phase. This study therefore aimed to investigate the impact of  
115 photo and thermooxidation processes on the biotic degradation of a pure uncrosslinked SBR sample,  
116 as a simpler system. Accelerated ageing studies were conducted to characterize the chemical  
117 modifications of the elastomer by these abiotic processes and quantify the degradation products  
118 formed by a multi analytical approach. The impact of abiotic degradation on biodegradation was  
119 assessed by monitoring the growth of two bacterial strains, belonging to two ubiquitous genera,  
120 *Rhodococcus ruber* ATCC 29672 and *Gordonia polyisoprenivorans* VH2 in the presence of pristine  
121 and variously oxidized SBR samples. The various sources of carbon (leachable and extractable  
122 compounds and SBR matrix) that could be potential substrates for microorganisms were monitored  
123 during biodegradation.

124

## 125 **2. Materials and methods**

### 126 *2.1. Chemicals and reagents*

127 All the solvents used for extraction were HPLC analytical grade. Acetone and tetrahydrofuran (THF)  
128 stabilized with 2,6-di-tert-butyl-4-methylphenol (THF stabilized with BHT, Analytical grade, q.s. 1  
129 g/L) were obtained from Sigma Aldrich. Deuterated water supplemented with 0.2 mM TSPd<sub>4</sub> (sodium  
130 tetra deuterated trimethylsilyl propionate) and acetone-d<sub>6</sub> supplemented with 0.03 v/v % TMS  
131 (tetramethylsilane) were purchased respectively from Eurisotop and InnovaChem.

132

### 133 *2.2. Sample preparation*

134 Two elastomers, provided by Michelin, were tested: a non-vulcanized styrene-butadiene rubber (SBR)  
135 containing 15 w% styrene, 20 w% 1,2-butadiene, 65 w% 1,4-butadiene (Mn ~210 kg.mol<sup>-1</sup>, Mw ~413  
136 kg.mol<sup>-1</sup>) and a synthetic *cis*-1,4-polyisoprene (IR) (Mn ~400 kg.mol<sup>-1</sup>). To remove antioxidants, SBR  
137 and IR, solubilized in toluene (solution at 100 mg.mL<sup>-1</sup> and 50 mg.mL<sup>-1</sup> for SBR and IR, respectively),  
138 were precipitated in 2 volumes of ethanol. The washing step was repeated on the coagulum twice. The  
139 elastomers were then vacuum-dried on an aluminum tray in an oven at 45 °C under nitrogen flux for  
140 about 40 h, down to a constant weight. The elastomers were then stored in sealed septum vials under  
141 inert atmosphere (nitrogen) in the dark at -25 °C.

142

### 143 *2.3. Abiotic degradation*

144 50 µm thick films were prepared by compression molding between silicone-coated polyester clothes  
145 (SRF121, HiFi Industrial Film), first in contact for 30 s at 110 °C and then pressed at 50 bars for 1 min  
146 at 110 °C. As the films obtained were not self-standing, they were aged directly on silicone-coated  
147 polyester clothes or placed onto CaF<sub>2</sub> pellets (Crystran) to facilitate further experiments.

148 Photoageing experiments were conducted using a Suntest CPS/XLS Atlas device, equipped with an  
149 ATLAS (NXE1700) xenon lamp set at 500 W.m<sup>-2</sup> in the UV-visible domain (300–800 nm) (Potaufoux

150 et al., 2022). Infrared irradiation and UV photons below 300 nm were cut off using an Atlas “daylight  
151 filter”. A cryostat was used to maintain the black standard temperature (BST) at 60 °C, corresponding  
152 to *ca.* 37 °C in the irradiation chamber.

153 Thermooxidative ageing was carried out in a forced air venting oven (Mettler UF30) at 80 °C.

154

#### 155 2.4. Biodegradation

156 *Preparation of the inoculum.* Two bacterial strains were selected for the biodegradation experiments:  
157 *Rhodococcus ruber* ATCC 29672 and *Gordonia polyisoprenivorans* VH2 (DSMZ 44266). They were  
158 cultivated at 27 °C for 24 h in shake flasks containing 10 mL of tryptic soy (TS) broth. Aliquots of  
159 both cultures (2 mL) were centrifuged at 9,700×g during 5 min and resuspended in NaCl solution (8%  
160 w/v). Pellets were washed two times more and finally resuspended in 2 mL of mineral salt medium  
161 (MSM). MSM was prepared as follow: Na<sub>2</sub>HPO<sub>4</sub>·12H<sub>2</sub>O, 3.8 g.L<sup>-1</sup>; KH<sub>2</sub>PO<sub>4</sub>, 1.8 g.L<sup>-1</sup>, MgSO<sub>4</sub>·7H<sub>2</sub>O,  
162 20 mg.L<sup>-1</sup>; Fe(NH<sub>4</sub>)<sub>2</sub>(SO<sub>4</sub>)<sub>2</sub>·6H<sub>2</sub>O, 30 mg.L<sup>-1</sup>; CaCl<sub>2</sub>·2H<sub>2</sub>O, 10 mg.L<sup>-1</sup>; NaCl, 0.5 g.L<sup>-1</sup>, NH<sub>4</sub>Cl, 0.3  
163 g.L<sup>-1</sup>; MnSO<sub>4</sub>·H<sub>2</sub>O, 18.5 mg.L<sup>-1</sup>, H<sub>3</sub>BO<sub>3</sub>, 20 mg.L<sup>-1</sup>; ZnSO<sub>4</sub>·7H<sub>2</sub>O, 2.5 mg.L<sup>-1</sup>; Na<sub>2</sub>MoO<sub>4</sub>·2H<sub>2</sub>O, 2.5  
164 mg.L<sup>-1</sup>; CoSO<sub>4</sub> 8 µg.L<sup>-1</sup> and CuSO<sub>4</sub>·5H<sub>2</sub>O 8 µg.L<sup>-1</sup>.

165 *Biodegradation experiments.* Biodegradation assays were performed in 50 mL sterile glass flasks.  
166 Strains were inoculated at 10<sup>6</sup> cell.mL<sup>-1</sup> from the previously prepared cell suspension in 10 mL of  
167 MSM containing or not (biotic control) 100±5 mg of SBR films as the sole carbon source. The films  
168 were previously sterilized, directly in the culture flasks, by soaking with 1 mL of 70 % ethanol until  
169 total evaporation of the ethanol. Bacteria were cultivated on four SBR materials: pristine SBR  
170 (“SBR”), photooxidized SBR (“SBR-photo-30h”) and two thermooxidized SBR (“SBR-thermo-168h”  
171 and “SBR-thermo-330h”). In parallel, bacteria were also cultivated on pristine IR in the same  
172 conditions for comparison. The condition *Gordonia*/IR was used as a positive control. Incubations of  
173 SBR and IR materials without bacteria were also performed as abiotic controls (AB). The experiments  
174 were carried out in duplicate for each condition, for 30 days at 27° C and 130 rpm in an Innova 44  
175 incubator (Eppendorf).



176 *Bacterial growth.* Biologic oxygen demand (BOD) of each culture was monitored using respirometric  
177 BOD Oxitop (WTW). Flasks were hermetically closed by Oxitop heads, which recorded BOD values  
178 every 120 min. BOD measurements were converted to percentage of biodegradation (% ThOD) by  
179 calculating the ratio of the oxygen consumption to the theoretical amount of consumed oxygen for  
180 complete degradation of the elastomer (ThOD) (Equation 1 and. 2, SI) according to the standard  
181 method ISO 14851 (ISO 14851:2019). At the end of the experiment, the growth of bacteria was  
182 determined by measuring the total ATP of culture using ATP Biomass Kit HS (Biothema).

183 *Extracellular protein content* (Bradford assay). At the end of the biodegradation experiment, a  
184 supernatant aliquot (500  $\mu$ L) was centrifuged at  $9,700 \times g$  during 5 min. to remove cells. In a  
185 transparent 96-well plate, 150  $\mu$ L of Bradford solution (ThermoFischer) were added to 150  $\mu$ L of the  
186 cell-free supernatant of each experiment. After 5 min of incubation, the plate was read at 595 nm in a  
187 Safire II<sup>®</sup> microplate reader (Tecan). Extracellular protein content was determined for each sample in  
188 duplicate according to the bovine serum albumin (BSA) calibration.

189

## 190 2.5. Sample characterizations.

191 From the pristine material, the abiotic and biotic degradation generate two distinct experimental series  
192 of samples comprising the residual film and the leachable and extractable short oxidized chains. For  
193 abiotic degradation samples, the films were characterized by Fourier Transform Infrared spectroscopy  
194 (FTIR), Gel fraction and the soluble fraction by size exclusion chromatography (SEC), upon the  
195 ageing experiments. The short oxidized chains were extracted from a 20 mg SBR film in 2 mL of  
196 sterile distilled water or acetone to distinguish the readily extractable (water-soluble) fraction from less  
197 soluble compounds (acetone-soluble). Extractions were performed in duplicate in sterile glass vials, on  
198 a horizontal shaker at 200 rpm and 25°C in an incubator (Bioconcept, Firlabo). The extracts were  
199 collected after 7 days of incubation. The water- and organo-soluble short oxidized chains were  
200 characterized by <sup>1</sup>H nuclear magnetic resonance spectroscopy (NMR), Liquid chromatography  
201 coupled with high resolution mass spectroscopy (HRMS) and Total organic carbon (TOC) analysis.

202 For biotic degradation samples, gel fraction analysis was performed on residual films. The water-  
203 soluble short oxidized chains were recovered from the supernatants of culture medium by  
204 centrifugation at  $9,700\times g$  for 5 min to remove cells. The SBR films (100 mg) were recovered from  
205 each culture and freeze-dried overnight. A 20 mg aliquot was extracted in 2 mL acetone for 7 days.  
206 The water- and acetone-soluble short oxidized chains were characterized by  $^1\text{H}$  NMR and LC/HRMS.

207

#### 208 2.5.1. Infrared analyses

209 Throughout the photo- or thermooxidative aging process, the kinetic characterization of chemical  
210 transformation was carried out by infrared analyzes on a single SBR film. Infrared spectra of SBR  
211 films coated on  $\text{CaF}_2$  were recorded in transmission mode with a Nicolet 6700 FTIR spectrometer,  
212 working with OMNIC software. Spectra were obtained using 32 scans and a  $4\text{ cm}^{-1}$  resolution. FTIR  
213 spectra obtained by attenuated total reflection mode (ATR) were recorded using a Nicolet 380  
214 spectrophotometer equipped with a diamond crystal (refractive index = 2.5). Spectra were recorded  
215 with 64 scans at a resolution of  $4\text{ cm}^{-1}$ .

216

#### 217 2.5.2. Size Exclusion Chromatography (SEC)

218 Following ageing experiments, degradation of SBR was also evaluated by characterizing the  
219 molecular weight distribution of the soluble fraction using SEC. SBR films aliquots ( $20 \pm 5\text{ mg}$ ) were  
220 dissolved in THF. The solutions were stirred for 2 h then filtered through  $0.45\text{ }\mu\text{m}$  disposable syringe  
221 filters (PTFE-45/25). The SEC equipment consisted of a degasser, injector and pump (Waters 2695E),  
222 a column heater (Waters, ALLCOLHTRB) and a refractive index (RI) detector (Waters, 2414). The  
223 columns were three PolyPore InfinityLab,  $7.5 \times 300\text{ mm}$  (Agilent, PL1113-6500). The columns were  
224 maintained at  $35^\circ\text{ C}$ . The mobile phase was THF at a flow rate of  $1.00\text{ mL}\cdot\text{min}^{-1}$ . The injected volume  
225 was  $100\text{ }\mu\text{L}$ . The signals obtained with SEC-RI detector were analyzed with Empower software  
226 (Waters) using Polystyrene (PS) calibration standards (PSS Polymer, PSS-pskitr1).

227

### 228 2.5.3. Gel fraction determination

229 To determine the gel fraction, soaking of SBR films (20 mg) was carried out during 72 h in 2 mL of  
230 THF at ambient temperature in the dark. Gel and soluble fractions were separated by vacuum filtration  
231 on glass fiber filters (Dutscher). The insoluble SBR (gel) was recovered on the filters and was  
232 weighted after vacuum drying (3 days at ambient temperature). Gel fractions (Gf in %) were  
233 determined by calculating the ratio of the mass of the insoluble SBR to the initial mass of SBR films  
234 according to the equation (1).

$$235 \quad (1) \text{ Gf (\%)} = [\text{SBR}]_{\text{gel}} / [\text{SBR}]_{\text{initial}} * 100$$

236

### 237 2.5.4. <sup>1</sup>H Nuclear Magnetic Resonance spectrometry

238 <sup>1</sup>H NMR was used to analyze the water, acetone and THF soluble short chains of SBR after abiotic  
239 and biotic degradation. The supernatants from the solvent extractions and biodegradation experiments  
240 (1200 μL) were dried under a nitrogen flux and 2-fold concentrated in 600 μL of the corresponding  
241 deuterated solvent (D<sub>2</sub>O or acetone-d<sub>6</sub> supplemented with a reference)). The NMR analyses were  
242 performed on a *Bruker* 500 MHz Avance III spectrometer and the acquisition method was detailed in  
243 Calarnou et al. (2023). Briefly, 8 scans were collected (3.24 s acquisition time, 60.0 s relaxation delay,  
244 a spectral window of 20 ppm, FID size of 65536 data points zero-filled to 131072 point).

245 The total NMR signals were integrated to assess the relative quantity of short oxidized chains in the  
246 samples and determine their production or consumption. To compare the samples, the total integration  
247 of the NMR signals was calibrated according to the internal deuterated standard - TSP for aqueous  
248 extracts and TMS for organic extracts (References value equal to 1). The short oxidized chain  
249 quantities were expressed in arbitrary units (a.u).

250

### 251 2.5.5. High resolution mass spectrometry

252 The supernatants (300  $\mu$ L) from solvent extractions and biodegradation experiments were dried under  
253 a nitrogen flux. Samples were 3-fold concentrated by dissolving them in 100  $\mu$ L of water / acetonitrile  
254 mixture (50:50 v:v) for water extracts, or 100  $\mu$ L of acetone for acetone extracts. HRMS analyses were  
255 performed with an Ultimate 3000 RSLC chromatographic system (ThermoScientific) coupled to an  
256 Exactive Orbitrap mass spectrometer (Thermo Fisher Scientific) equipped with an electrospray (ESI)  
257 source and the chromatographic separation was detailed in Calarnou et al. (2023). The mass  
258 spectrometer was operated in positive ion mode and the detection was performed with full scan from  
259 m/z 50 to 750 using a resolution set at 70 000 at m/z 200. HRMS raw data were processed with  
260 Xcalibur Software (version 4.1). For each sample, all the m/z values detected in the chromatographic  
261 run with a peak intensity strictly greater than  $10^4$  were considered. The molecular formulae were  
262 determined by the software algorithm from the exact masses of the m/z (with a mass tolerance of 3  
263 ppm). Due to the chemical nature of SBR and its potential degradation products, single charged ions  
264 and only carbon, hydrogen, oxygen atoms and single sodium adducts were considered in the  
265 calculation with the following range of atoms: C (1–100), H (1–200), O (0–50), Na (0–1). For each  
266 feature, the molecular formula was used to determine the number of carbon, hydrogen and oxygen  
267 atoms, and to calculate the ratios of H/C and O/C. The complex molecular composition of the samples  
268 was presented using Van Krevelen diagrams which consist in plotting the H/C ratio against the O/C  
269 ratio (Kim et al., 2003; Van Krevelen, 1950). All the plots were generated using the open-source  
270 software pyC2MC (Sueur et al., 2023). To monitor the biodegradation of low-molecular-weight  
271 compounds resulting from the abiotic degradation, only the m/z values also present in the abiotic  
272 control sample were considered. In the biotic sample data, these values were distinguished and  
273 extracted from those of the biomass using a Python script.

274

#### 275 2.5.6. Total organic carbon (TOC)

276 The supernatants (600  $\mu$ L) of water and acetone extractions prepared after abiotic degradation were  
277 dried under a nitrogen flux. The sample duplicates were pooled in 8 mL of ultrapure water before the  
278 measurement. The concentration of the total organic carbon (TOC) was determined using a Shimadzu  
279 TOC 5050A analyzer.

280

### 281 **3. Results and discussions**

#### 282 *3.1. Abiotic degradation*

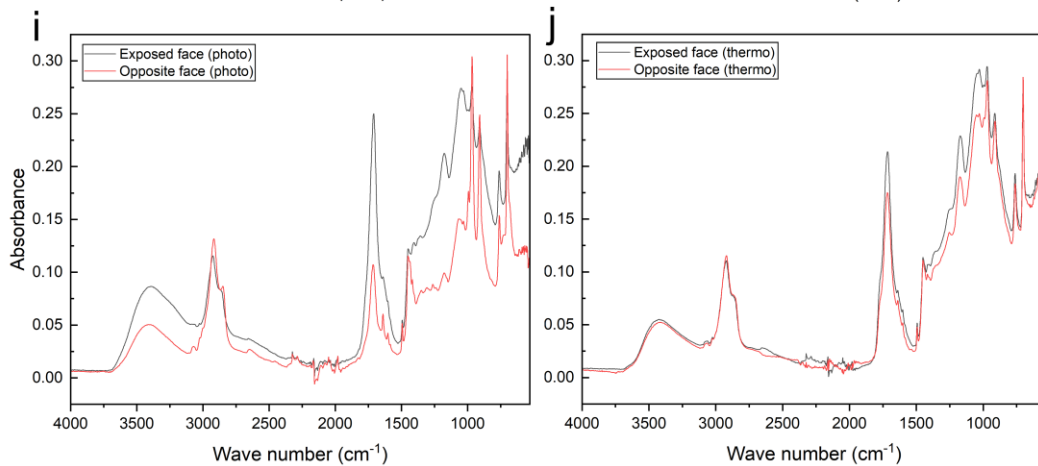
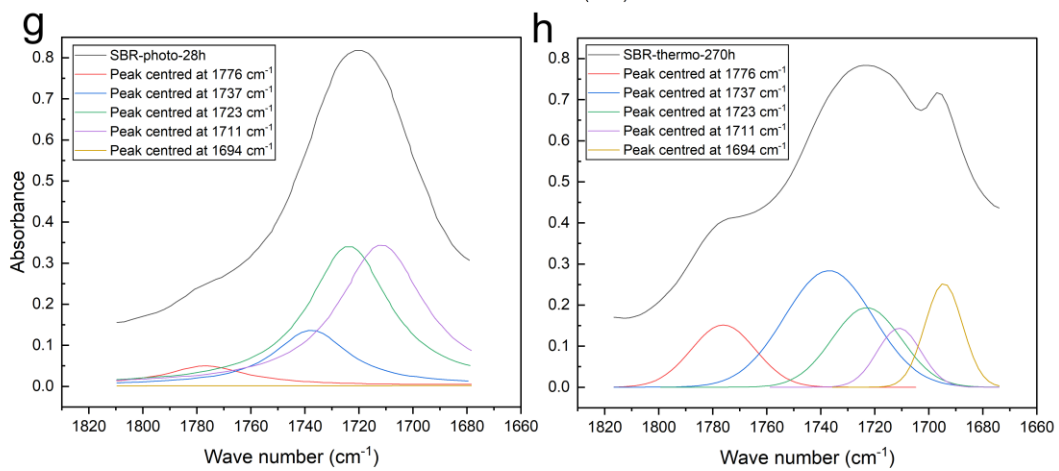
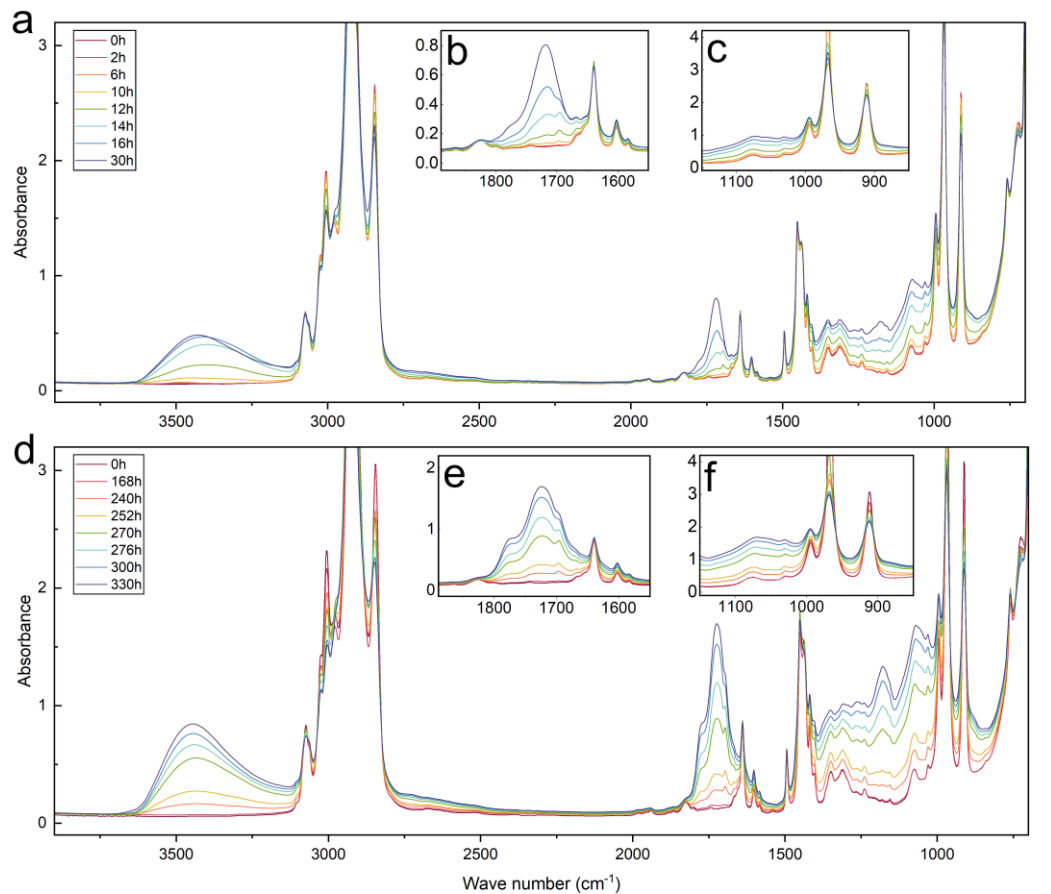
283 Abiotic photochemical- and thermal-ageing of SBR were assessed first by performing kinetic  
284 measurements to characterize the gradual chemical transformations of the elastomer in terms of  
285 oxidation, crosslinking, and chain scission.

286 The FTIR spectrum of a SBR film showed the characteristic bands of styrene (1456, 1495,  
287 1582, 1601  $\text{cm}^{-1}$ ) and of 1-4 and 1-2 butadiene units with the vibrations of methylene groups in the  
288 sequence 1,4-*trans* and 1-2, at 966  $\text{cm}^{-1}$  and 912  $\text{cm}^{-1}$ , respectively (Fig. S1).

289

##### 290 *3.1.1. Chemical modifications resulting from photooxidation*

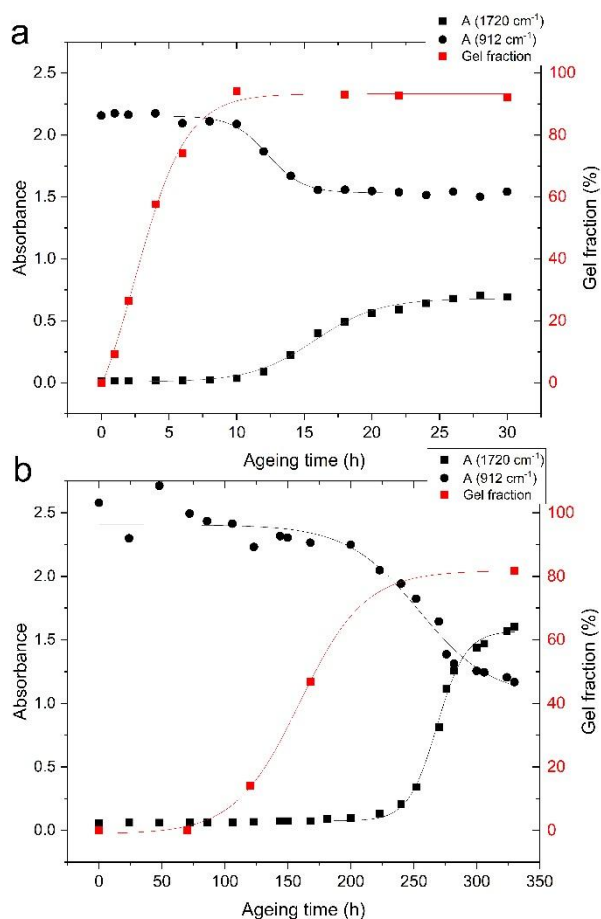
291 The FTIR spectra (transmission mode) of SBR films exposed to long wavelength irradiation  
292 ( $\lambda > 300$  nm) with oxygen presented various modifications from photooxidation reactions (Fig. 1a, 1b,  
293 1c).



295 **Fig. 1. a-f)** Transmission FTIR spectra of a SBR film ( $e = 50 \mu\text{m}$ ) (onto  $\text{CaF}_2$ ) **a-c)** photooxidised at  
296  $\lambda > 300 \text{ nm}$ ,  $37^\circ \text{C}$ : a) between  $4000\text{-}800 \text{ cm}^{-1}$ , b) in the carbonyl/carboxyl domain  $1900\text{-}1550 \text{ cm}^{-1}$ , c)  
297 in the domain  $1150\text{-}850 \text{ cm}^{-1}$ ; **d-f)** during thermooxidation at  $80^\circ \text{C}$ : d) between  $4000\text{-}800 \text{ cm}^{-1}$ , e) in  
298 the carbonyl domain  $1900\text{-}1550 \text{ cm}^{-1}$ , e) in the domain  $1150\text{-}850 \text{ cm}^{-1}$ . **g-h)** Deconvoluted FTIR  
299 spectra after 28 h of photooxidation (g) or 270 h of thermooxidation (h). **i-j)** ATR-FTIR spectra of the  
300 front face (exposed to oxygen) and opposite face for a SBR film onto silicone-coated polyester clothes  
301 i) after 30 h of photooxidation j) after 330h of thermooxidation at  $80^\circ \text{C}$ .

302

303 Spectral modifications, presented in Fig. 1a to 1c were identified from previous studies  
304 performed using chemical derivatization treatments and physical treatments (photolysis, wavelengths  
305 effects) (Adam et al., 1990, 1989b, 1989a; Jouan et al., 1989; Wilhelm and Gardette, 1994). In the  
306 hydroxyl region ( $3600\text{-}3200 \text{ cm}^{-1}$ ), a broad absorption band appeared with a maximum around  $3460$   
307  $\text{cm}^{-1}$  corresponding to hydroperoxide and alcohol formation. The carbonyl region in Fig. 1b showed a  
308 broadband with a maximum at  $1720 \text{ cm}^{-1}$  that results from the convolution of two bands ( $1711$  and  
309  $1723 \text{ cm}^{-1}$ ), and a shoulder to this broadband appeared at  $1775 \text{ cm}^{-1}$ . In the first 10 h of irradiation, an  
310 absorption band was observed at  $1694 \text{ cm}^{-1}$  attributed to  $\alpha\beta$ -unsaturated ketones, (Adam et al., 1990).  
311 The latter are rapidly transformed by addition of radicals to give saturated ketones at  $1723 \text{ cm}^{-1}$ .  
312 Ketones absorb at wavelengths longer than  $300 \text{ nm}$  and react through Norrish reactions to produce  
313 carboxylic acids at  $1711 \text{ cm}^{-1}$  (Fig. S2). Kinetic monitoring of the absorption band intensities is  
314 presented in Fig. 2a, which shows the increase of absorbance of the carbonyl/carboxyl band at  $1720$   
315  $\text{cm}^{-1}$  and the decrease of that of unsaturation at  $912 \text{ cm}^{-1}$  (the absorbance at  $966 \text{ cm}^{-1}$  is too high and  
316 cannot be used in transmission mode for a  $50 \mu\text{m}$  thick SBR film).



317

318 **Fig. 2.** Variations of absorbance at  $1720\text{ cm}^{-1}$  (carbonyl/carboxyl;  $\blacksquare$ ),  $912\text{ cm}^{-1}$  (vinyl;  $\bullet$ ) and gel  
 319 fraction level ( $\blacksquare$ ) as a function of ageing time for SBR films during (a) photooxidation at  $\lambda > 300$  nm  
 320 and (b) thermooxidation at  $80^\circ$  C.

321

322 The shape of the kinetic curves in Fig. 2a indicated that the decrease of the absorbance of the  
 323 vinyl unsaturation (at  $912\text{ cm}^{-1}$ ) was inversely proportional to the formation of carbonyls. A period  
 324 with no detectable oxidation of SBR was observed during the first 10 h of irradiation. Then, the  
 325 absorbance of the carbonyls (at  $1720\text{ cm}^{-1}$ ) increased between 10 h and 25 h and reached a plateau  
 326 with an absorbance around 0.7 (film thickness  $50\text{ }\mu\text{m}$ ). At the same time, the absorbance of vinyl  
 327 double bond decreased to 32 % of their initial absorbance up to a plateau corresponding to an  
 328 absorbance of 1.5. The formation of carbonylated photoproducts and the consumption of double bonds



329 reached a steady state and photooxidation stopped after about 25 h of irradiation. The styrene  
330 absorption bands (1456, 1495, 1582, 1601  $\text{cm}^{-1}$ ) did not show any significant changes.

331

### 332 3.1.2. Chemical modifications resulting from thermooxidation

333 The FTIR spectra of SBR films (Fig. 1d, 1e, 1f) submitted to thermal ageing presented  
334 modifications like those observed in photooxidation.

335 However, the relative intensities of the absorption bands in the carbonyl domain were  
336 different. The band at 1694  $\text{cm}^{-1}$  increased along thermooxidation, and the saturated ketone absorption  
337 band (1723  $\text{cm}^{-1}$ ) prevailed on the carboxylic acid absorption band at 1711  $\text{cm}^{-1}$ . This is a consequence  
338 of Norrish reactions, which only occur in photooxidation. The differences between photo- and  
339 thermooxidation are highlighted in Fig. 1g and 1h, which represent the deconvoluted spectra after 28 h  
340 of photooxidation and 270 h of thermooxidation.

341 As shown in Fig. 2a and 2b, different levels of oxidation were reached in thermo- and  
342 photooxidation. These results suggest that the bulk oxidation was higher under thermooxidative  
343 conditions than in photooxidative conditions, which reflects an oxidation profile in the case of  
344 photooxidation, confirmed when comparing the ATR-FTIR spectra of the front and rear surfaces of  
345 samples irradiated for 30 h or thermooxidized for 330 h (Fig. 1 i-j). In the case of thermooxidation, the  
346 rear side was practically as oxidized as the front, whereas in the case of photooxidation, it was  
347 significantly less oxidized. This heterogeneous oxidation results from a limited oxidation due to an  
348 oxygen starvation effect. A higher level of oxidation was observed in photooxidation. The surface  
349 oxidation is an important parameter to consider for biodegradation, as it will be colonized by  
350 microorganisms and will be one of the starting points of microbial attack.

351 The observed decrease of unsaturation (vinyl at 912  $\text{cm}^{-1}$ , Fig. 2) was less for photooxidation.  
352 One should also consider that in the case of photooxidation, the photolysis of the ketones by Norrish  
353 type II reaction provokes the formation of vinyl double bonds (912  $\text{cm}^{-1}$ ). The loss of vinyl  
354 unsaturation is then, at least partially, compensated for by their formation.

355

### 356 *3.1.3. Crosslinking and chain scission reactions*

357           Besides the production of chemical moieties, oxidation reactions generate new chemical bonds  
358 or chain scissions. After each ageing time, the determination of gel fraction revealed the insoluble  
359 portion of SBR rapidly increased over the first 10 h of irradiation leading to a plateau of about 95 %  
360 (Fig. 2a). This increase can be evidenced even before the detection of any oxidation products by FTIR  
361 spectroscopy. In the case of thermooxidation, the gel fraction rose after 70 h and reached progressively  
362 a plateau of 80 % of insoluble after 300 h of ageing (Fig. 2b). Formation of this insoluble part reflects  
363 the creation of a network SBR structure through crosslinking reactions.

364           In addition, a decrease in the molecular weight of the samples was measured by size exclusion  
365 chromatography in the resulting soluble fraction, showing the occurrence of chain scissions under both  
366 conditions (Tables S1 and S2). These results indicated that the ratio of crosslinks to chain scissions are  
367 different depending on the ageing conditions and help us to understand the results of FTIR  
368 characterization. It was indeed observed that, as a result of photooxidation, the surface layers were  
369 quickly and strongly crosslinked, which decreased oxygen permeability, as previously reported by  
370 Adam et al. (Adam et al., 1990, 1989b). This contributed to the limitation of the photooxidation by the  
371 diffusion of oxygen, even in the case of this study where the thickness of the aged films is relatively  
372 weak ( $e=50\ \mu\text{m}$ ). Second, the absorbance of carbonylated products was limited to a plateau value of  
373 only 0.7. On the other hand, in the case of thermooxidation, with less crosslinking, the oxygen  
374 permeability is still sufficient to allow the bulk layer to be as oxidized as the front layer, as shown in  
375 Fig. 1h. The maximal value reached for the carbonyl absorbance measured by transmission  
376 spectroscopy was also much higher than for photooxidation. Our results and hypotheses agree with  
377 previous studies showing a decrease of the oxygen diffusion coefficient during the oxidation of SBR  
378 and polybutadiene, consistent with reduced oxygen permeability due to the rubber crosslinking  
379 (Clough and Gillen, 1992; Mertz et al., 2012; Sinturel and Billingham, 2000).

380 In any case, the oxidation processes induced significant transformations in the SBR network  
381 resulting in the formation of new covalent bonds in the degraded network, low molecular weight and  
382 highly functionalized products resulting from chain scissions. However, we are unable to discuss the  
383 environmental representativity of our results relative to a SBR from a full TRWP composition, as the  
384 oxidation state and crosslinking degree of SBR chains network within TRWP have not been reported  
385 yet. It can be expected that oxidation and scission/crosslinking reactions of vulcanized SBR in tire  
386 particles also occur in the environment (Cadle and Williams, 1980; Wagner et al., 2022). Indeed, the  
387 increase of the crosslinking density was already observed for unfilled SBR vulcanizates after photo-  
388 ageing (Karekar et al., 2023) and for carbon-black filled SBR vulcanizates after thermo-ageing (Rezig  
389 et al., 2020) but the characterization of short oxidized chains of SBR has never been reported.

390

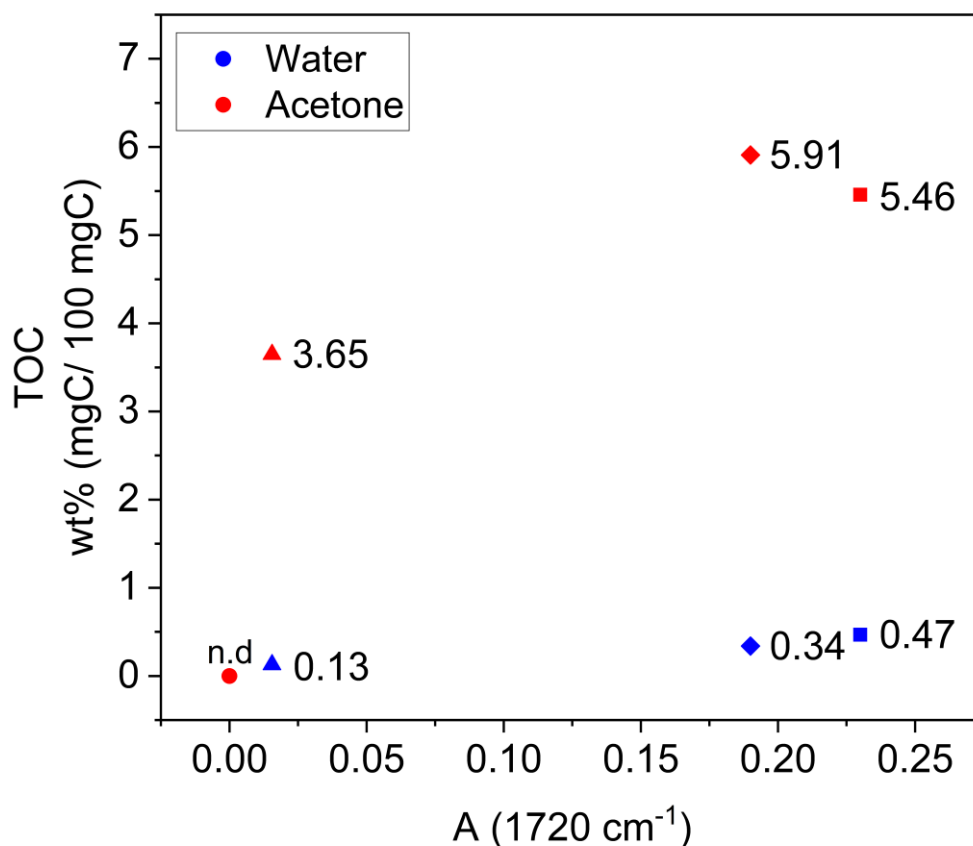
#### 391 *3.1.4. Extraction and characterization of SBR short, oxidized chains*

392 Short, oxidized chains resulting from chain scissions upon photo and thermooxidation could  
393 then be further leached out in aqueous solution as water-soluble products or remained in the films  
394 (surface and bulk) as extractable compounds. Pristine, 30 h photooxidized (SBR-photo-30h), 168 h  
395 and 330 h thermooxidized (SBR-thermo-168h and SBR-thermo-330h) SBR films were extracted in  
396 water and acetone to characterize qualitatively and quantitatively the leachable and extractable  
397 compounds, respectively.

##### 398 *3.1.4.1. Total organic carbon*

399 Total organic carbon analysis was performed to evaluate the amount of such short chains  
400 released in the water and acetone and was expressed as a function of the oxidation level measured by  
401 ATR-FTIR at the surface of the sample ( $1720\text{ cm}^{-1}$ ). Our results indicated that the amount of carbon in  
402 these extracts increased as a function of the surface oxidation level of SBR (Fig. 3). As expected, for  
403 pristine SBR, no extractable short, oxidized chain was detected either in water or in acetone (below the  
404 detection limit). For SBR-thermo-168h (low oxidized surface), short, oxidized chains were released as  
405 a small proportion (0.13 % and 3.65 % of total carbon in water and acetone, respectively). In contrast,

406 for the more oxidized samples (SBR-thermo-330h and SBR-photo-30h), a higher amount of carbon  
407 was released, in acetone especially. Proportions for photo and thermooxidation are equivalent.  
408 Leached products of SBR-thermo-330h and SBR-photo-30h represented 0.34 % and 0.47 % of the  
409 total carbon, respectively, whereas higher proportions (5.91 % and 5.46 %, respectively) were released  
410 in acetone extractable fractions. These results confirm the production of numerous short, oxidized  
411 chains under oxidative conditions and corroborate the decrease of molecular weight previously  
412 observed. Such water leachable and acetone extractable short chains could represent bioavailable  
413 substrates for bacteria. The combined fractions account for approximately 6 % of the total carbon (6  
414 mg TOC out of 100 mg of the initial SBR-thermo-330h and SBR-photo-30h). The amount of short,  
415 oxidized chains was systematically higher in the acetone extract than in water indicating that such  
416 products constitute an intermediate pool of carbon between water soluble products and the insoluble  
417 fraction.



418

419 **Fig. 3.** Proportion of water (blue symbols) and acetone (red symbols) extracted short oxidized chains  
420 from SBR samples after 7-day extraction (wt% in mgC/ 100 mgC): pristine SBR (●), SBR-thermo-  
421 168h (▲), SBR-thermo-330h (◆) and SBR-photo-30h (■), determined by TOC measurements as a  
422 function of absorbance at  $1720\text{ cm}^{-1}$  (carbonyl groups) at the surface of the films (FTIR-ATR). No  
423 carbon content was detected for the pristine SBR (n.d, below the detection limit).

424

#### 425 3.1.4.2. $^1\text{H}$ NMR

426 The characterization of the short, oxidized chains resulting from chain scission and their  
427 quantification was complemented by  $^1\text{H}$  NMR analysis. Specific profiles were obtained for each water  
428 and acetone extract of SBR sample and full spectrum integration was used to assess their relative  
429 abundance (Fig. S3). The results confirm the trends previously observed with TOC analyses showing  
430 an increase of the total amount of short, oxidized chains as a function of oxidation extent.

431 Besides this quantitative analysis, the  $^1\text{H}$  NMR spectra provide a qualitative characterization  
432 of the different chemical structures of the SBR short, oxidized chains. Characteristic regions of the  
433 spectra were assigned to styrene aromatic protons ( $\delta_{\text{H}} \sim 7.0\text{-}7.7$  ppm in water;  $\delta_{\text{H}} \sim 6.8\text{-}7.7$  ppm in  
434 acetone), butadiene ethylenic protons ( $\delta_{\text{H}} \sim 5.2\text{-}6.2$  ppm in water;  $\delta_{\text{H}} \sim 4.7\text{-}5.9$  ppm in acetone) and  
435 aliphatic structures ( $\delta_{\text{H}} \sim 0.7\text{-}3.2$  ppm in water;  $\delta_{\text{H}} \sim 0.7\text{-}2.8$  ppm in acetone) (Fig. S3 and S4).  
436 Notably, the extracts exhibit high signals in the region of hydroxylated groups -OH ( $\delta_{\text{H}} \sim 3.2\text{-}4.7$  ppm)  
437 and low signals in the region of aldehyde protons —CHO ( $\delta_{\text{H}} \sim 9.3\text{-}9.8$  ppm in water;  $\delta_{\text{H}} \sim 9.3\text{-}9.9$   
438 ppm in acetone), confirming that both leachable and extractable short chains contained oxidized  
439 moieties (Fig. S4). Structural assignment revealed more butadiene ethylenic protons in acetone  
440 extracts (20-30 % of total signal) than in water extracts (>10 % of total signal), confirming the  
441 extraction of longer aliphatic chains. On the contrary, the protons related to hydroxylated groups,  
442 represented 20 % and less than 10 % of total signal of water and acetone extracts, respectively. These  
443 results indicated that after abiotic degradation, short, oxidized chains can be released into water and

444 longer oxidized chains remained in the SBR films, as expected considering the effect of chain length  
445 on polarity and water solubility.

446 While studies on the characterization of leachable compounds are still scarce and mainly focus on  
447 chemical additives (Müller et al., 2022; Seuront et al., 2022), note that liquid  $^1\text{H}$  NMR appears as a  
448 convenient tool to monitor the short oxidized chains that can be released after abiotic degradation, as  
449 already described for polyethylene (Eyheraguibel et al., 2017), polypropylene (Fontanella et al., 2013)  
450 or more recently polystyrene and polyethylene terephthalate (Giaganini et al., 2023). These short,  
451 oxidized chains produced and released in water can be considered as potential carbon substrates for  
452 microorganisms in the environment.

453

#### 454 3.1.4.3. High resolution mass spectrometry

455 The characterization of the short, oxidized chains was finally completed with LC/HRMS  
456 analysis (examples of chromatograms are given in Fig. S5). For all samples, mass spectra showed a  
457 broad distribution of peaks in a mass range between 50 to 750  $\text{g}\cdot\text{mol}^{-1}$  (Fig. S6) but the composition of  
458 each spectrum varied in number of peaks and intensity. The water extracts were characterized by  
459 complex mass spectra consisting of hundreds of peaks (SBR-thermo-168h: 1645 peaks, SBR-thermo-  
460 330h: 1894 peaks, SBR-photo-30h: 1768 peaks) (Table 1). In the acetone extracts, a similar number of  
461 peaks was found for SBR-thermo-168h (1606 peaks), while 2400 peaks were detected for the high  
462 oxidized SBR-thermo-330h and 2523 for SBR-photo-30h. This large number of features reflected the  
463 diversity of the short, oxidized chains resulting from the oxidation and chain scission processes  
464 occurring during photo and thermooxidation. The extracts were characterized by an average molecular  
465 weight ( $M_w$ ) of 326 and 352  $\text{g}\cdot\text{mol}^{-1}$  in water and up to 407 and 412  $\text{g}\cdot\text{mol}^{-1}$  in the acetone extracts for  
466 SBR-photo-30h and SBR-thermo-330h, respectively (Table 1). Previous studies on the abiotic  
467 degradation of polyethylene already reported similar molecular weight ranges and number of peaks,  
468 indicating that abiotic oxidation leads to the production of small compounds lower than 1000 Da  
469 (Eyheraguibel et al., 2018, 2017). These short oxidized chains must be considered as potential

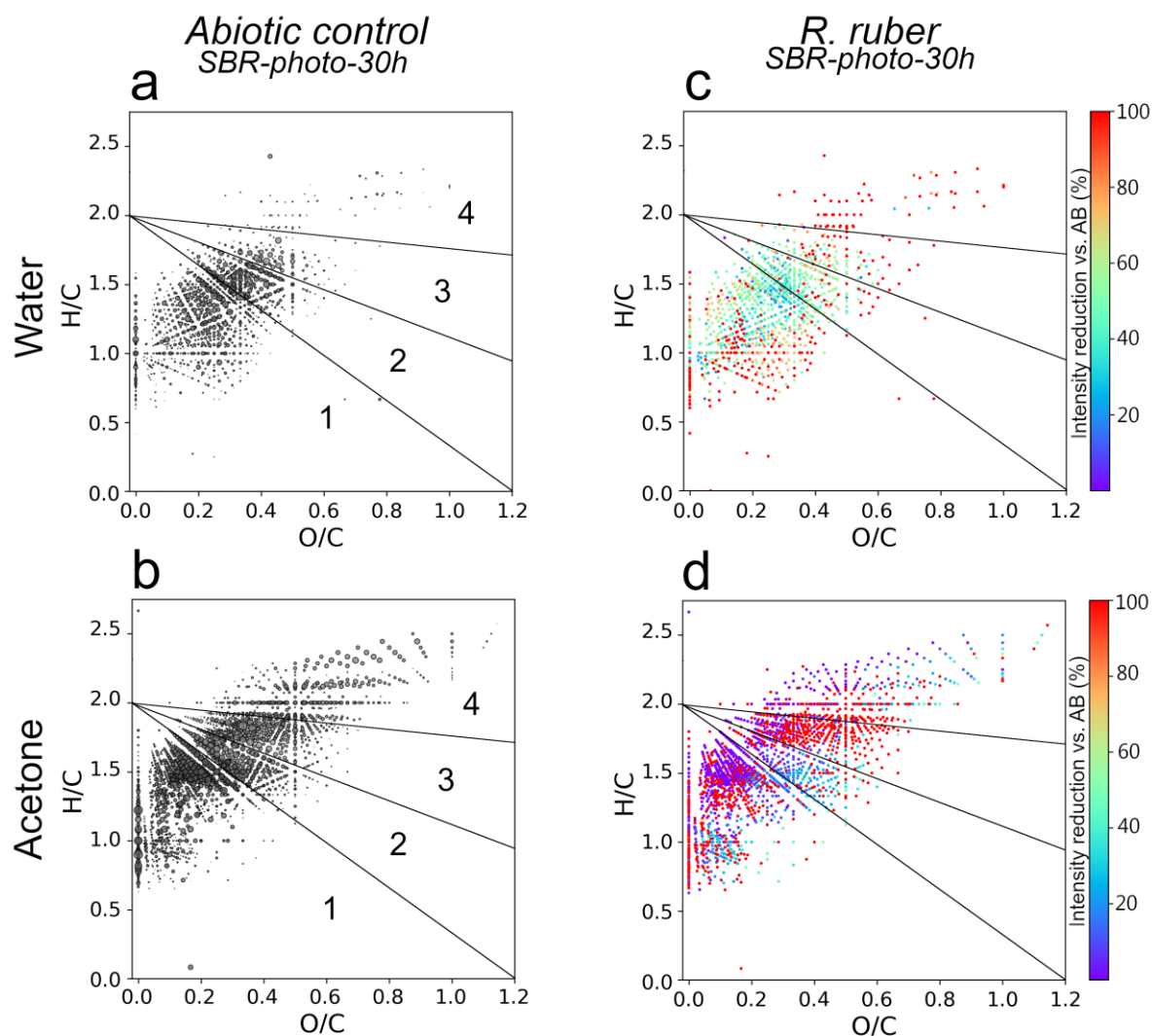
470 substrates in addition to the uncrosslinked and crosslinked SBR chains resulting from abiotic ageing.  
471 So far, very few studies focused on SBR degradation products. Only recently, Thomas et al. (2022)  
472 proposed a list of possible transformation products of elastomer resulting from thermal or photo-  
473 ageing of tire cryogrinds. Oxidation products such as carboxylic acids, ketones and epoxides were  
474 detected in THF extract by GC/MS, but few compounds could be identified through targeted approach  
475 as no database provides a comprehensive list of degradation products. In this study, Van Krevelen  
476 diagrams (VKD) were used to better visualize and understand the complex chemical composition of  
477 the extracts (Fig. 4a and 4b). Specific regions of the diagram provide information on alkylation and  
478 oxidation of the compounds. The most oxidized compounds have high O/C ratios ( $> 0.5$ ) and appear  
479 on the right side of the plot. Aliphatic compounds typically have high H/C ratios ( $> 1.5$ ) and low O/C  
480 ratios ( $< 0.5$ ) and aromatic hydrocarbons have low H/C ratios ( $< 1.0$ ) and O/C ratios ( $< 0.5$ ) and  
481 appear respectively in the upper left and lower left part of the diagram (Kim et al., 2003; Kourtchev et  
482 al., 2014; Noziere et al., 2015). On the Fig. 4a and 4b, four regions were defined to represent (1)  
483 highly unsaturated compounds, (2) intermediately oxidized, (3) highly oxidized and (4) very highly  
484 oxidized according to Sueur et al. (2023). The distribution of the short, oxidized chains was similar for  
485 both ageing and extracting conditions with a wider distribution of compounds in acetone extract (Fig.  
486 4b) corresponding to a higher number of peaks detected. The VKD showed that random chain  
487 scissions occurred producing mixed short, oxidized chains containing all the aromatic/unsaturated,  
488 aliphatic, and oxidized moieties present originally in the chemical structure of the SBR monomers  
489 (Fig. S4). The most represented compounds contained unsaturated moieties (region 1) attributed to  
490 styrene and diene bonds. Intermediately oxidized compounds (region 2) were present in both water  
491 and acetone extracts in large quantity while highly and very highly oxidized compounds (regions 3  
492 and 4) appeared with a larger amount in acetone extract, reflecting the advanced oxidation level of the  
493 surface and bulk film. Further work will be required to identify these short, oxidized chains.

494

495 **Table 1.** Characteristics of SBR water (supernatant) and acetone extracted short, oxidized chains and  
 496 residue after 30-day incubation with *R. ruber* (RR), *G. polyisoprenivorans* (GP) or without bacteria  
 497 (abiotic control (AB)). nd = non-detected. The mean values ( $n=2 \pm SD$ ) of gel fraction are presented.

Materials		SBR	SBR-thermo-168h			SBR-thermo-330h			SBR-photo-30h		
Sample			AB	RR	GP	AB	RR	GP	AB	RR	GP
Number of peaks	Water	nd	1645	654	1035	1894	1500	1511	1768	1105	1550
	Acetone	nd	1606	1007	1355	2400	2113	1389	2523	2390	1488
Mw (g.mol <sup>-1</sup> )	Water	nd	323	317	344	352	341	352	326	335	346
	Acetone	nd	333	387	416	412	413	418	407	401	406
Gf	%	0	35±7	19±2	44±3	80±4	87±3	89±9	81±1	89±1	88±3

498



499



500 **Fig. 4.** Van Krevelen diagrams of LC/MS ESI (+) data showing **on the left column**, the H/C vs. O/C  
501 atomic ratio for compounds in the water (a) and acetone (b) extract of the SBR-photo-30h abiotic  
502 control (AB). Point size represents the peak intensity. They are separated in 4 regions: (1) highly  
503 unsaturated compounds, (2) intermediately oxidized, (3) highly oxidized and (4) very highly oxidized;  
504 **on the right column**, the degradation of compounds (peak intensity reduction (%)) in the water  
505 (supernatant) (c) and acetone (d) extract of SBR-photo-30h after a 30-day incubation with *R. ruber*.  
506 Intensities above  $10^4$  were considered. The level of degradation is represented by a color scale from  
507 fully degraded (red) to not degraded (purple).

508

### 509 3.2. Biotic degradation

#### 510 3.2.1. Bacterial growth

511

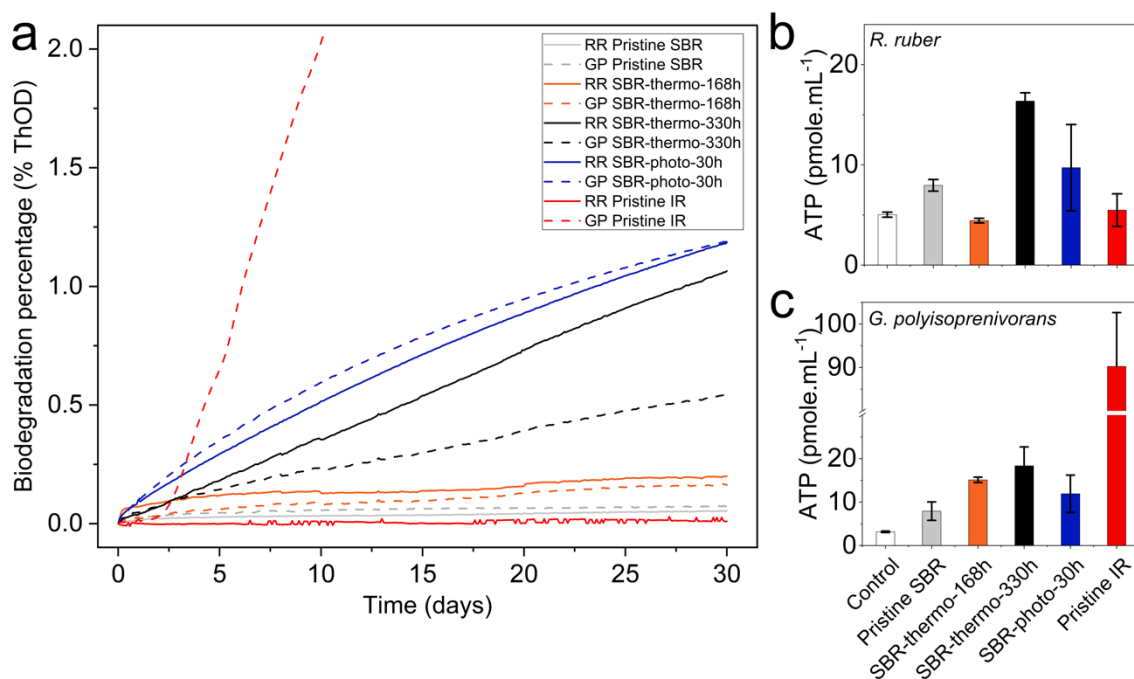
512 The biodegradation of pristine and photo- or thermooxidized SBR films by two bacterial  
513 strains, *R. ruber* and *G. polyisoprenivorans*, was investigated. For a given SBR sample, similar growth  
514 was observed with both strains, but the rate of biodegradation (% ThOD, SI equation 1) increased with  
515 the level of SBR oxidation (Fig. 5a). The lowest biodegradation percentage was observed for pristine  
516 SBR (less than 0.1 % for both strains) and a slight increase was observed for SBR-thermo-168h (0.2  
517 %) after 30 days of incubation. Higher biodegradation percentages were obtained with more oxidized  
518 samples, up to 0.5-1 % with SBR-thermo-330h and up to 1.2 % with SBR-photo-30h. For each pre-  
519 oxidized SBR, the bacterial growth started immediately and kept rising for 30 days without reaching a  
520 plateau. Although growth curves for both strains were similar for most SBR samples, a significant  
521 difference was observed for SBR-thermo-330h, showing a higher biodegradation capacity of *R. ruber*  
522 in this case. The measurement of the ATP content confirmed these trends and clearly showed the  
523 production of more biomass with pristine and oxidized SBR, compared to the control, reflecting the  
524 variable bacterial growth (Fig. 5b). A slight increase of ATP amount was measured for pristine SBR  
525 suggesting a minimal bacterial growth within the timeframe of the experiment. The amount of ATP  
526 was much higher when bacteria were cultivated on more oxidized samples, except for the condition *R.*

527 *ruber* / SBR-thermo-168h. Globally, the oxidation (photo and thermo) of SBR promoted both  
528 mineralization and biomass production.

529 Both strains were isolated from the environment and appeared as newly identified degraders of  
530 photo and thermooxidized SBR. The biodegradation capacity of *R. ruber* has already been reported in  
531 the literature on various pre-oxidized polymers such as polyethylene (Bonhomme et al., 2003;  
532 Eyheraguibel et al., 2017), polypropylene (Fontanella et al., 2013) and more recently, extractable SBR  
533 from environmental aged roadway particles (Calarnou et al., 2023). Our results confirmed that *R.*  
534 *ruber* can degrade pre-oxidized SBR as the sole carbon source. So far, there is no evidence of bacterial  
535 growth on similar oxidized SBR samples. Only a few studies reported bacterial isolates able to grow  
536 on vulcanized SBR (ground tire rubber) such as *Acidithiobacillus ferrooxidans*, *Gordonia* sp. and  
537 *Sphingomonas* sp., in a context of devulcanization and recycling (Cui et al., 2016; Li et al., 2013).  
538 Tsuchii et al. also reported that unvulcanized SBR and BR were hardly attacked by *Nocardia* sp. and  
539 *Moraxella* sp. (Tsuchii et al., 1985, 1984). The authors indicated that only BR samples with an  
540 average molecular weight of 2,500 or less were degraded and assimilated by *Moraxella* sp. in 5 days  
541 (Tsuchii et al., 1984). This observation supports our results which showed that the low molecular  
542 weight generated in pre-oxidized SBR samples (soluble fraction of THF) might enhance  
543 biodegradation.

544 In our study, complementary control experiments carried out on IR substrate provided  
545 different results for the two strains tested. The biodegradation percentage obtained with *G.*  
546 *polyisoprenivorans* on IR reached 3% within the first 20 days and levelled off until the end of the  
547 experiment probably due to the limiting oxygen in the system (Fig. S7). The biomass production  
548 (ATP) exceeded the level of the biotic control 28-fold, and at least 5-fold that of oxidized SBR  
549 samples, exhibiting significantly higher growth on IR for this strain (Fig. 5c). Such results were  
550 expected as *G. polyisoprenivorans* VH2 is a well-known bacterium able to degrade IR with high  
551 biodegradation rates (20 % of mineralization in 50 days) (Andler et al., 2018; Linos and Steinbüchel,  
552 1998; Warneke et al., 2007). Interestingly, they also point out that *G. polyisoprenivorans* biodegrade  
553 pristine SBR, pristine IR and oxidized SBR at various rates and suggest that different mechanisms of

554 degradation could occur according to the material. Moreover, no increase in bacterial growth or ATP  
 555 content was recorded for *R. ruber* with IR, suggesting that metabolic pathways involved in the  
 556 biodegradation of oxidized SBR were not effective for the degradation of IR. These results emphasize  
 557 the capacity of each strain to grow on aged SBR and the need to perform further research to elucidate  
 558 their mechanisms of biodegradation.



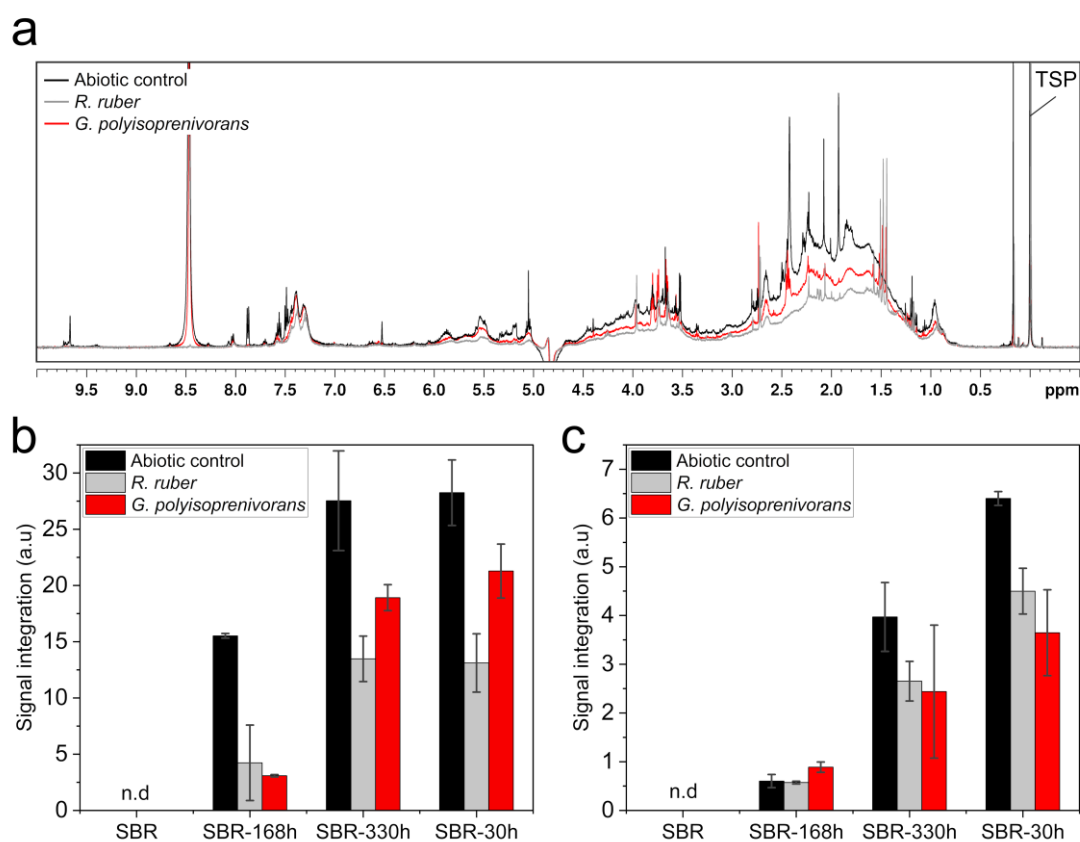
559  
 560 **Fig. 5.** Bacterial growth (biodegradation kinetics (a, % ThOD) and total ATP content (b, c) of *R.*  
 561 *ruber* (solid lines, b) and *G. polyisoprenivorans* (dashed lines, c) cultivated on pristine SBR (grey),  
 562 SBR-thermo-168h (orange), SBR-thermo-330h (black), SBR-photo-30h (blue) and pristine IR (red) or  
 563 without any carbon source (Control) for 30 days. The mean values (n=2 ± SD) are presented.

564

### 565 3.2.2. Evaluation of biodegradation by <sup>1</sup>H NMR spectroscopy

566 At the end of the biodegradation experiment (30 days of incubation), the consumption of short,  
 567 oxidized chains as potential bioavailable carbon sources was assessed by comparing the <sup>1</sup>H NMR  
 568 signal intensities of each sample to the abiotic control (Fig. 6a). The water extractable products  
 569 recorded a 25-50 % decrease of the signal intensity for SBR-thermo-330h and SBR-photo-30h

570 samples and reached more than 75 % for the SBR-thermo-168h after 30 days of culture (Fig. 6b).  
 571 Similarly, the signal intensity of acetone extracts decreased from 30 % to 43 % for SBR-photo-30h  
 572 and SBR-thermo-330h samples but no decrease was observed for SBR-thermo-168h (Fig. 6c).  
 573 Moreover, no significant difference in the relative proportion of the previously assigned region was  
 574 observed, suggesting that removal (either by bacterial uptake or enzymatic transformation) did not  
 575 target any preferential chemical structure (Fig. S8, S9 and S10). The results suggested the  
 576 bioassimilation of the short, oxidized chains instead, and indicated that the strains can use both  
 577 compounds leached out in the supernatant and the extractables remaining in the oxidized films. The  
 578 consumption of the acetone extracted compounds could suggest an adhesive growth of both bacteria.  
 579 Indeed, the *Rhodococcus* and *Gordonia* genera were already reported as adhesive bacteria on rubber  
 580 substrates (Basik et al., 2021; Chengalroyen and Dabbs, 2013) and their development at the surface of  
 581 rubber was already observed by scanning electron microscopy (SEM) (Andler et al., 2022; Berekaa et  
 582 al., 2000).



583

584 **Fig. 6. (a)**  $^1\text{H}$  NMR spectra of the SBR-photo-30h supernatants obtained with *R. ruber* (grey), *G.*  
585 *polyisoprenivorans* (red) and without bacteria (abiotic control) (black) after 30-day incubation. For a  
586 better visualization, only one replicate is presented on the figure. NMR relative integrations (arbitrary  
587 unit) of water-soluble (supernatant) **(b)** and acetone **(c)** extracted short, oxidized chains with *R. ruber*  
588 (grey bars), *G. polyisoprenivorans* (red bars) or without bacteria (abiotic control) (black bars). No  
589 signal was detected for the pristine SBR (n.d). Arbitrary units (a.u) are not comparable between water  
590 and acetone analyses as the internal standard used was different. The mean values ( $n=2 \pm \text{SD}$ ) are  
591 presented.

592

### 593 3.2.3. Evaluation of biodegradation by mass spectrometry

594 After the biodegradation experiment, the number of peaks detected by mass spectrometry decreased in  
595 different proportions according to the SBR extract analyzed (Table 1), revealing that 6 % to 60 % of  
596 compounds were completely biodegraded. Van Krevelen diagrams were plotted using a color scale  
597 from red (fully degraded) to purple (no degradation) to monitor the evolution of the short, oxidized  
598 chain composition in the water (supernatants) and acetone extracts of SBR-photo-30h (Fig. 4c-d and  
599 S11), SBR-thermo-330h (Fig. S12 and S13) and SBR-thermo-168h (Fig. S14 and S15). In the  
600 presence of *R. ruber* and *G. polyisoprenivorans*, numerous compounds were highly or partly degraded  
601 in both water and acetone extracts (red dots). This confirmed that biodegradation of short, oxidized  
602 chains can occur in the medium and at the surface of the films. The average molecular weight of  
603 remaining short, oxidized chains did not change indicating the degradation of compounds with sizes  
604 from 50 up to  $750 \text{ g.mol}^{-1}$  (Table 1). Moreover, all the different classes of compounds were impacted  
605 showing that the chemical structure of the oxidized chains did not limit biodegradation. However,  
606 aromatic compounds (region 1 - H/C ratios  $<1.0$ ) seemed more degraded than aliphatic ones (region 1  
607 - H/C ratios  $>1.5$ ) while highly oxidized molecules (regions 3 and 4) were strongly degraded (Fig. 4c  
608 and 4d). These results did not allow us to conclude that there is a preferential degradation of a  
609 chemical structure.

610

#### 611 3.2.4. Evolution of the insoluble fraction of SBR

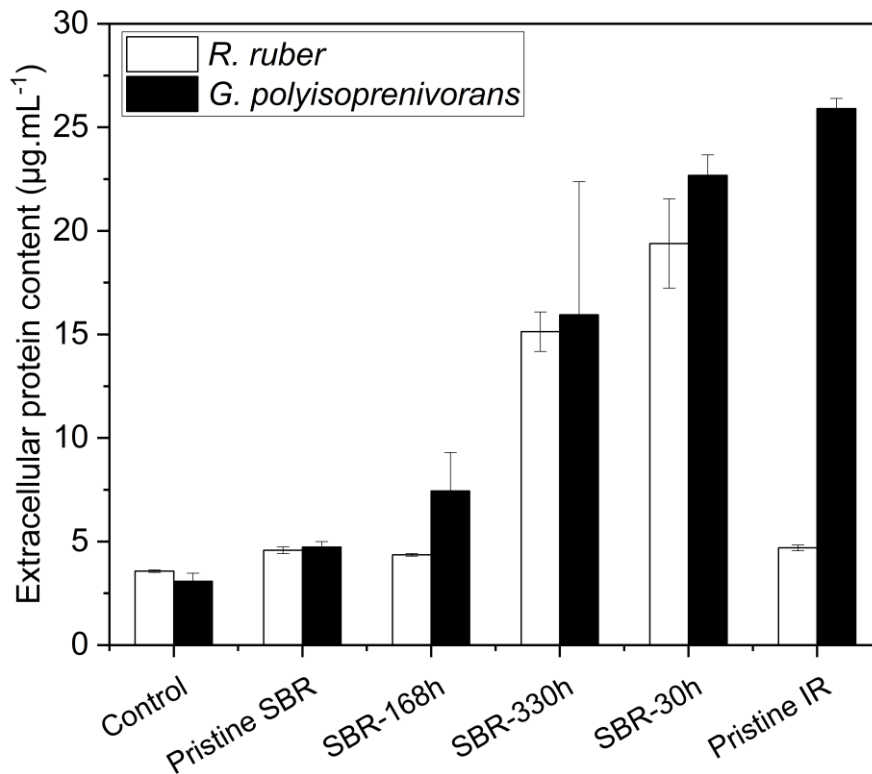
612 After abiotic degradation, the insoluble crosslinked SBR fraction and THF soluble fraction could also  
613 be a potential carbon resource for bacteria as it represented 35 % of the carbon of SBR-thermo-168h  
614 samples and 80 % on average for the most oxidized SBR-thermo-330h and SBR-photo-30h samples  
615 (Table 1). At the end of biodegradation experiments, the insoluble fraction of SBR-thermo-330h and  
616 SBR-photo-30h tends to increase by 10 % on average in the presence of both strains compared to the  
617 abiotic control (Table 1), suggesting a concomitant reduction of the THF soluble fraction. This  
618 increase even reached 25 % for SBR-thermo-168h with *G. polyisoprenivorans*. Combined with the  
619 mass spectrometry data, these results might suggest that no or low degradation or cleavage of the  
620 insoluble fraction occurred within the 30 days of culture and longer incubation times must be tested to  
621 conclude on the use of the crosslinked network as a carbon substrate. As expected, the short oxidized  
622 chains biodegraded faster than the high molecular weight crosslinked chains, as reported for BR  
623 (Tsuchii et al., 1984). On the other hand, the THF soluble fraction of the residual film was clearly  
624 degraded by the two strains, indicating that biodegradation processes take place on several carbon  
625 sources. Conversely, the SBR-thermo-168h insoluble fraction percentage with *R. ruber* is almost  
626 halved. This surprising result could suggest a partial attack of the insoluble residue as moreover, a  
627 high biodegradation level of the soluble short, oxidized chain compounds was also observed with this  
628 strain (-60 % and -37 % of the number of peaks detected by mass spectrometry in the supernatant and  
629 in the acetone extract after biodegradation, respectively, (Table 1) with a high biodegradation  
630 percentage (Fig. S14). Nevertheless, these results do not match with those obtained for biological  
631 parameters, the BOD-based biodegradation percentage and ATP content for *R. ruber* remaining low  
632 (Fig. 5). As for the other samples, longer incubation time must be tested to conclude on the use of the  
633 crosslinked network as a carbon substrate.

634

#### 635 3.2.5. Extracellular protein content

636 Previous works have shown the key role of extracellular enzymes to attack polymer surface and  
637 released oligomers (Chow et al., 2022; Hiessl et al., 2012; Jendrossek, 2019; Odobel et al., 2021;

638 Santo et al., 2013). Therefore, protein amount in the supernatant was measured for the different  
639 samples. The protein content increased with the level of oxidation of the SBR samples. The lowest  
640 protein content ( $6 \mu\text{g.mL}^{-1}$ ) was observed in SBR-thermo-168h supernatant while higher contents were  
641 measured in the case of SBR-thermo-330h ( $16 \mu\text{g.mL}^{-1}$ ) and SBR-photo-30h ( $20\text{-}23 \mu\text{g.mL}^{-1}$ ) for both  
642 strains (Fig. 7). Similarly, the protein content was positively correlated with the level of consumption  
643 of short, oxidized chains. It would now be interesting to assess whether these proteins correspond to  
644 enzymes potentially involved in the degradation of SBR material or the cleavage of oxidized chains to  
645 enhance their assimilation. To date, strong focus was more on extracellular oxidative enzymes  
646 efficient towards IR and NR. Indeed, some enzymes were reported in the literature to cleave C=C  
647 bonds of polyisoprene such as peroxidase and laccase (Cheng et al., 2023), latex clearing protein  
648 (Hiessl et al., 2014), rubber oxygenases RoxA and RoxB (Birke et al., 2017; Jendrossek and  
649 Reinhardt, 2003). In our positive control experiment, a strong growth of *G. polyisoprenivorans* was  
650 observed on IR and the extracellular protein measurement reached  $26 \mu\text{g.mL}^{-1}$  (Fig. 7) probably  
651 reflecting the production of Lcp1<sub>VH2</sub> and Lcp2<sub>VH2</sub> enzymes as described in the literature (Andler, 2020;  
652 Andler and Steinbüchel, 2017; Hiessl et al., 2014). Low extracellular protein contents were detected in  
653 the presence of pristine SBR, suggesting an absence of enzymatic activity on the non-oxidized SBR.  
654 On the contrary, high level of extracellular proteins ( $23 \mu\text{g.mL}^{-1}$ ) was measured in the supernatant of  
655 *G. polyisoprenivorans* growing on oxidized SBR. but further analyses are required to confirm the  
656 presence of enzymes and test their potential activity on oxidized SBR.



657

658 **Fig. 7.** Extracellular protein content (Bradford assay) in supernatants of *R. ruber* and *G.*  
 659 *polyisoprenivorans* cultivated 30 days in the presence of pristine SBR, SBR-thermo-168h, SBR-  
 660 thermo-330h, SBR-photo-30h or pristine IR. The mean values ( $n=2 \pm SD$ ) are presented.

661

### 662 3.3 Limitation of environmental model research: pristine SBR degradation to represent TRWP?

663 As a major component of tire tread, SBR undergoes various stresses during the tire use phase and after  
 664 its emission into the environment in TRWP. The purpose of this work was to study a simplified system  
 665 (pristine SBR) to overcome the complexity of TRWP composition and focus on the mechanisms  
 666 underlying the degradation of SBR. This first step highlighted that the aging of a pure SBR induced  
 667 significant transformations of the elastomer via oxidation processes. The formation of new covalent  
 668 bonds resulted in a crosslinked SBR network, while chain scissions concurrently led to the formation  
 669 of low molecular weight and highly functionalized products, that can then be biodegraded.  
 670 Nevertheless, our model system will need to be progressively complexified to more realistically mimic



671 the fate of the elastomer phase within TRWP in an environmental context and to understand what will  
672 impact its biodegradation: vulcanization, elastomer microstructure and the presence of other  
673 components embedded in TRWP (fillers, plasticizers, protective and vulcanizing agents) should  
674 especially be considered for future studies.

675

#### 676 4. Conclusion

677 The present study provides new insights about the abiotic degradation of SBR and its impact on  
678 biodegradation. Under abiotic conditions, photo or thermo-oxidation processes lead to the production  
679 of multiple compounds, mainly short, oxidized chains resulting from main chain scissions and  
680 containing high amount of unsaturation and carbonylated functions. These short chains can be released  
681 in the water phase to become potential substrates for microorganisms. Further investigations are now  
682 required to identify their chemical composition.

683 The capacity of two strains, *Rhodococcus ruber* and *Gordonia polyisoprenivorans*, to grow on  
684 weathered SBR material was demonstrated for the first time. Different carbon sources were considered  
685 for biodegradation such as the short, oxidized chains and the crosslinked matrix resulting from abiotic  
686 degradation. Our results indicate that both strains were able to degrade the low molecular weight  
687 oxidized SBR but not the pristine SBR. These results evidence the positive impact of abiotic  
688 degradation on the biodegradation process. It would now be interesting to check whether this  
689 biodegradation requires the action of specific bacteria such as *Rhodococcus ruber* and *Gordonia*  
690 *polyisoprenivorans*, or could be observed with other strains or microbial consortia. The high  
691 extracellular protein content, measured when the strains were grown on highly oxidized SBR samples,  
692 raised questions about the potential involvement of enzymatic activities toward the SBR material or its  
693 abiotic degradation product (short, oxidized chains). Further investigations are in progress in our  
694 laboratory.

695

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698

699

700 **References**

701

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